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# Compressional behavior of omphacite to 47 GPa

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Abstract Omphacite is an important mineral component of eclogite. Single-crystal synchrotron X-ray diffraction data on natural (Ca, Na) (Mg, Fe, Al)Si<sub>2</sub>O<sub>6</sub> omphacite have been collected at the Advanced Photon Source beamlines 13-BM-C and 13-ID-D up to 47 GPa at ambient temperature. Unit cell parameter and crystal structure refinements were carried out to constrain the isothermal equation of state and compression mechanism. The third-order Birch-Murnaghan equation of state (BM3) fit of all data gives  $V_0 = 423.9(3)$  Å<sup>3</sup>,  $K_{T0} = 116(2)$  GPa and  $K_{T0}' = 4.3(2)$ . These elastic parameters are consistent with the general trend of the diopside-jadeite join. The eight-coordinated polyhedra (M2 and M21) are the most compressible and contribute to majority of the unit cell compression, while the SiO<sub>4</sub> tetrahedra (Si1 and Si2) behave as rigid structural units and are the most incompressible. Axial compressibilities are determined by fitting linearized BM3 equation of state to pressure dependences of unit cell parameters. Throughout the investigated pressure range, the b-axis is more compressible than the *c*-axis. The axial compressibility of the *a*-axis is the largest among the three axes at

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0 GPa, yet it quickly drops to the smallest at pressures above 5 GPa, which is explained by the rotation of the stiffest major compression axis toward the a-axis with the increase in pressure.

**Keywords** Pyroxene · High pressure · Single-crystal diffraction · Synchrotron · Mantle · Subduction

# Introduction

Eclogite is a high-pressure mantle metamorphic rock, and it is a major constituent of the subducted slab, forming mainly from basalt or gabbro at depths greater than 40 km (corresponding to pressures greater than 1.2 GPa). Containing only minor fraction of olivine, eclogite is denser than surrounding mantle materials and provides one of the primary driving forces of the subduction process (Anderson 2007; Moghadam et al. 2010). Depending on the exact buoyancy conditions determined by the temperature profile of the subduction zone, slabs that reach to the mantle transition zone may stagnate and be subhorizontally deflected (Fukao et al. 2009; King et al. 2015; van Mierlo et al. 2013); however, in some subduction zones slabs are able to penetrate the transition zone and sink to the lower mantle (Hirose et al. 1999; van der Hilst and Karason 1999). Effects of metastable preservation of some of the upper mantle minerals including olivine and pyroxenes in the cold subduction environments on the slab buoyancy have recently been demonstrated to have significant implications for the slab stagnation (Agrusta et al. 2014; King et al. 2015; van Mierlo et al. 2013). A good understanding of the compression behavior and stability limits of mineral components of eclogite at high pressures helps to constrain the force balance of the subduction and provides valuable

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information for modeling of the dynamics of subduction zones (Moghadam et al. 2010). In addition to anomalously high density, eclogite is also characterized by thermal conductivity lower than that of typical mantle harzburgite (Wang et al. 2014) and thus may serve as thermal insulating layer for the slab and control the heat flow.

Eclogites are usually composed of two major mineral phases: omphacite [(Ca, Na)(Mg, Fe, Al)Si<sub>2</sub>O<sub>6</sub>] and garnet (McNamara 2012; Moghadam et al. 2010). Among these two minerals, omphacite commonly serves as the framework mineral of the eclogite assembly, while garnet is dispersed in various morphologies throughout the omphacite framework (McNamara 2012). Microstructures of eclogites found at the Earth's surface show that omphacite controls the bulk rheological properties of eclogites at natural conditions (McNamara 2012; Moghadam et al. 2010). Undeformed garnet grains are usually embedded in a continuous omphacite matrix with preferred orientation (Moghadam et al. 2010). Experiments on the highpressure behavior of natural omphacite provide therefore important information constraining the rheological behavior of eclogite at mantle conditions. Omphacite is also responsible for the low thermal conductivity of eclogite (Wang et al. 2014). In addition to eclogitic xenoliths, omphacites are often found as solid inclusions in kimberlitic diamonds (Mitchell and Giardini 1977; Promprated et al. 2004; Smyth 1980; Sobolev et al. 1999) and have been reported in chondritic meteorite samples (Kimura et al. 2013). While nominally anhydrous, natural omphacites often exhibit significant level of hydration, with OH concentrations reaching as high as 700 ppm (Koch-Muller et al. 2004, 2007), and are expected to be one of the significant water carriers into the mantle.

Depending on the crystallization conditions, omphacites can assume one of two different crystal structures: the substitutionally disordered high-temperature phase with space group C2/c and the cation-ordered phase with space group P2/n (Fleet et al. 1978; McNamara 2012; Moghadam et al. 2010; Oberti and Caporuscio 1991). At ambient pressure, the order-disorder transition takes place at 725 °C (Fleet et al. 1978). Temperature quench rate controls the diffusion-driven redistribution of cations on cooling, allowing to preserve the substitution-disordered state to ambient conditions on rapid cooling. In general, omphacites from tectonic environments, in which exhumation proceeds slowly, tend to be of the P2/n variety, whereas specimens from kimberlitic or meteoritic origins usually exhibit the disordered symmetry. Because of this temperature sensitivity of cation distribution, omphacites are used as a base of the omphacite-garnet geothermometer (Råheim and Green 1974). The information about exact thermoelastic properties of omphacite is also valuable for geobarometric interpretation of elastic equilibria in host-diamond-inclusion systems to constrain residual pressure and formation conditions of the inclusions (Liu et al. 1990; Zhang 1998).

From a chemical composition perspective, omphacite can be considered as an intermediate phase in diopside– jadeite solid solution system. In the chemically diverse pyroxene family, simple linear mixing models for predicting physical properties of solid solutions usually work quite well; however, in omphacite, the complicated cation ordering effects make this approach inadequate. For example, thermal conductivity (Wang et al. 2014) and single-crystal elasticity (Skelton and Walker 2015) of P2/n omphacite have been shown to significantly depart from the linear trends. Similar nonlinear mixing effects have been demonstrated in spodumene (Sondergeld et al. 2006), which also exhibits order–disorder polymorphism.

Omphacite like other members of the pyroxene family is not expected to be thermodynamically stable above 25 GPa; however, recent results on the kinetics of pyroxene-garnet reaction (Nishi et al. 2008; van Mierlo et al. 2013), as well as experiments confirming metastable preservation of pyroxenes to pressures as high as 50 GPa (Dera et al. 2013a; Finkelstein et al. 2015; Hu et al. 2015), at lower temperatures strongly suggest that omphacite may be present in cold subducted slabs past the mantle transition zone. Limited pressure range of both experimental and computational studies on omphacite at high pressure conducted thus far makes it difficult and unreliable to extrapolate the physical properties to these extreme conditions, particularly considering that pyroxenes are known to exhibit subtle discontinuities of their compressional behavior in the lower pressure range (Dera et al. 2013a; Finkelstein et al. 2015). To properly account for this metastable component of eclogite in geodynamic models, it is therefore important to constrain the compressional behavior of omphacite to much higher pressures. In this paper, we present results of a compressional single-crystal synchrotron X-ray diffraction study of a natural-ordered P2/n omphacite up to 47 GPa at ambient temperature.

### **Experiments**

The natural (Ca,Na)(Mg,Fe,Al)Si<sub>2</sub>O<sub>6</sub> omphacite sample measured in this study comes from an eclogite rock fragment found in a locality near Occidental in Sonoma County, CA, and was obtained from the University of Arizona RRUFF collection (#R061129). Electron microprobe analysis conducted at the University of Arizona determined the chemical composition of the sample as (Ca<sub>0.51</sub>Na<sub>0.48</sub>) (Mg<sub>0.44</sub>Al<sub>0.44</sub>Fe<sup>2+</sup><sub>0.14</sub>Fe<sup>3+</sup><sub>0.02</sub>)Si<sub>2</sub>O<sub>6</sub>. The composition of the sample expressed in end-member molar percentages is: Di<sub>44</sub>Jd<sub>44</sub>Hd<sub>7</sub>Ae<sub>4</sub>Fs<sub>1</sub>. For simplicity, hereafter we will refer to the sample as Di<sub>51</sub>Jd<sub>49</sub>. One chip of omphacite single

crystal of about 20  $\mu$ m  $\times$  20  $\mu$ m  $\times$  5  $\mu$ m in size was used for this experiment. The omphacite sample was first mounted onto a polymer micromesh sample holder (MiTe-Gen) for room pressure measurement. Then, the same crystal was loaded into a BX90 DAC (Kantor et al. 2012). The BX90 DAC maximizes the opening angle ( $\sim \pm 30^{\circ}$  in our case), which is preferable for the single-crystal diffraction study. Two type I diamonds with culets of 300 µm in diameter were mounted on Boehler-Almax-type WC seats. A rhenium gasket was pre-indented to ~30 µm thickness, and a 180-µm-diameter hole was drilled in the center of the pre-indention using a laser milling machine. The omphacite crystal was loaded into the center of the gasket hole, and then, the sample chamber was loaded with He as the pressure-transmitting medium using the GSECARS gasloading system (Rivers et al. 2008). After He loading, the gasket hole shrank to about 100 µm in diameter. Throughout the whole experiment, despite gradual collapse of the gasket hole, the sample crystal stayed clear of the Re gasket edge. A few ruby spheres (~10  $\mu$ m in diameter) were placed around the omphacite sample to determine the pressure in the sample chamber (Mao et al. 1986). At each pressure, ruby fluorescence spectra were taken both before and after the diffraction data collection, and the average pressure values are reported here. All the diffraction patterns were collected at ambient temperature.

Single-crystal X-ray diffraction at ambient pressure was carried out at the experimental station 13-BM-C of the Advanced Photon Source, Argonne National Laboratory. The X-ray beam was monochromated with silicon 311 crystal to 28.6 keV (0.434 A), with 1 eV bandwidth. A Kirkpatrick-Baez mirror system was used to obtain a vertical  $\times$  horizontal focus spot size of 15  $\mu$ m  $\times$  15  $\mu$ m, measured as full width at half maximum (FWHM). The MAR165 charge-coupled device (CCD) detector (Ravonix) was placed about 175 mm away from the sample, and ambient LaB<sub>6</sub> powder was used to calibrate the distance and tilting of the detector. The sample was placed at the rotation center of the diffractometer and was aligned with optical microscope. A wide rotation exposure covered angular range from  $\varphi = -90^{\circ}$  to 90°, followed by a series of step  $\varphi$ -exposures, each covering 1° scan width. The typical exposure time was 1 s/°. At zero diffractometer position, the  $\varphi$  scan rotation axis for the 13-BM-C diffractometer is in the horizontal plane of the instrument and is perpendicular to the incident X-ray direction. After the first set of wide and step  $\varphi$  exposures, collected at zero detector position, similar data were recorded with the detector rotated about the horizontal axis (2 $\theta$ ) by 20° and then with the detector rotated about the vertical axis ( $\nu$ ) by 20°. The wide rotation exposures were used to extract d-spacings, azimuthal angles around the beam center, peak intensities of each diffraction peak and the step  $\varphi$  exposures provided the third spatial coordinate necessary for reconstructing the crystal's reciprocal lattice and were used to index the diffraction pattern.

High-pressure single-crystal X-ray diffraction images were collected at the neighboring experimental station 13-ID-D. The X-ray was monochromated to 37.0 keV (0.3344 A) with a silicon 111 crystal and then focused to  $4 \,\mu\text{m} \times 4 \,\mu\text{m}$  at the FWHM with a pair of K–B mirrors. The same MAR165 CCD detector was placed ~200 mm away from the sample in the DAC, and the detector was also calibrated with LaB<sub>6</sub> powder at ambient conditions. At each pressure, an  $\omega$ -scan with X-ray diode was collected to determine the maximum opening angle and to determine the functional shape of the absorption effect of the diamond anvils. After the  $\omega$ -scan, a wide  $\omega$  exposure with typical exposure time of  $1 \text{ s/}^{\circ}$  was carried to cover the maximum opening angle that the DAC allowed, followed by a series of step  $\omega$  exposure, each covering 1° scan width. The  $\omega$  scan rotation axis was vertical and perpendicular to the incident X-ray direction. In the 13-ID-D station, multiple detector positions were achieved by translating the detector horizontally by  $\pm 70$  mm, whereas in the 13-BM-C station, a detector rotation arm was used instead. The exposure times for the wide  $\omega$  rotation images were doubled when the detector was at off-center position because peaks at higher scattering angle were characterized by lower intensities.

The diffraction images were analyzed using the ATREX IDL software package (Dera et al. 2013b). Polarization, Lorentz and empirically determined diamond absorption corrections were applied to the fit peaks. The unit cell and orientation matrix were determined in RSV for each dataset. Lattice parameters were refined in RSV using a least squares fitting procedure. Changes in the sample illuminated volume and the absorption effects are corrected using the ATREX software (Görbitz 1999; Dera et al. 2013b). Corrected peak intensities were used to refine the crystal structures with SHELXL software, facilitated by WINGX and Olex2 general user interface (Dolomanov et al. 2009; Farrugia 2012; Sheldrick 2008). The crystal structure model from Mottana et al. (1979) was used as the initial model in the refinement. We used isotropic atomic displacement parameters (Uisos) for all atoms. The following restraints were used in the structural refinement: (1) all structural sites were fully occupied, and the site occupancy factors were allowed to vary freely if more than one cation occupied the same site; (2) Ca<sup>2+</sup> and Na<sup>2+</sup> were distributed between M1 and M11, and there were no other cations occupying these two sites; (3) M2 site was only occupied by Mg<sup>2+</sup> and Fe cations, and M21 site was only occupied by Al<sup>3+</sup> and Fe cations; (4) Si1 and Si2 site only contained  $Si^{4+}$  cation; (5) site occupation factors for  $Si^{4+}$  and  $O^{2-}$  sites were set to 1; (6) cations which occupy the same polyhedral site (M1, M11, M2 and M21) shared the same  $U_{iso}$  value; (7) all  $U_{iso}$  values are positive. At some pressures, the  $U_{iso}$ s of similar sites (e.g., M1/M11, M2/M21, Si1/Si2) showed strong trade-offs during fitting, and the  $U_{iso}$ s of some sites could be negative if allowed to fit freely. Given this condition, the  $U_{iso}$ s of similar sites was fitted dependently as the same value, which reduced the number of fitting parameters (supplementary materials Table 1). Refinement details, lattice parameters, fractional coordinates, atomic displacement parameters and bond lengths at different pressures are given in supplementary materials Tables 1–4. We used the VESTA software (Momma and Izumi 2008) to calculate the polyhedral volumes, average bond lengths and distortions, which are given in supplementary materials Table 5.

# **Results and discussions**

#### Equation of state of omphacite

Throughout the investigated pressure range, the omphacite sample retains the P2/n structure. Our P–V dataset is consistent with previous studies on P2/n omphacite with similar compositions (Fig. 1). The isothermal bulk modulus at 0 GPa,  $K_{T0}$ , and its pressure derivative,  $K_{T0}'$ , were determined by fitting the pressure–volume relation with a third-order Birch–Murnaghan equation of state (BM3) (Angel 2000). The data were fit using EoSFit 7c software (Angel et al. 2014), and the fitting was weighted by the uncertainties of both pressure and volume. The best fit gives



Fig. 1 Pressure dependence of unit cell volume of  $Di_{51}Jd_{49}$  and the best-fit BM3 equation of state. *Error bars* are smaller than the symbols



Fig. 2 Isothermal bulk moduli and their pressure derivatives of different clinopyroxenes along the diopside-jadeite join



**Fig. 3**  $f_E$ - $F_E$  plots of all data (*blue circles*) and the high-pressure data only (red diamonds). *Straight lines* demonstrate the best-fit BM3 equations of state, which are weighted linear fits of the  $f_E$ - $F_E$  datasets

 $V_0 = 423.9(3)$  Å<sup>3</sup>,  $K_{T0} = 116(2)$  GPa and  $K_{T0}' = 4.3(2)$ . Since the ambient pressure data and the high-pressure data were collected with different instrumentations, to check whether the equation of state fitting was biased by mixing the data, we also fitted the equation of state with only the high-pressure data, which were collected with the same instrumentation. The fitting results are illustrated as "HP only" in Figs. 1, 2 and 3. Best fit of the HP only data gives  $V_0 = 422.7(4)$  Å<sup>3</sup>,  $K_{T0} = 122(3)$  GPa and  $K_{T0}' = 4.0(2)$ . The two equations of state are nearly identical throughout the pressure range investigated (Fig. 1). On the other hand, we would like to note here that the ambient pressure data point has a better accuracy, because it does not have a limitation of opening angle and therefore has a better coverage of diffraction peaks, so we believe that the equation of state fitted with all data should represent the compressional behavior of this omphacite sample better. In Fig. 2, we compare the bulk modulus and its pressure derivative from this study with the same quantities reported previously (Gavrilenko et al. 2010; Mccarthy et al. 2008; Nestola et al. 2006; Pandolfo et al. 2012; Pavese et al. 2001; Zhang et al. 1997). Along the diopside-jadeite join, the isothermal bulk modulus  $(K_{TD})$ increases from ~105 GPa (diopside, Gavrilenko et al. 2010; Zhang et al. 1997) to ~135 GPa (jadeite, McCarthy et al. 2008; Nestola et al. 2006), and the  $K_{T0}$  of the P2/n omphacite has an intermediate value (115-122 GPa). As a reference, the ambient adiabatic bulk modulus of the C2/c omphacite polymorph determined by means of Brillouin spectroscopy was 130.8(5) (Bhagat et al. 1992).  $K_{70}$  is negatively correlated with  $K_{T0}$ .  $K_{T0}'$  decreases from ~6 (diopside, Gavrilenko et al. 2010; Zhang et al. 1997) to ~3.5 (jadeite, Mccarthy et al. 2008; Nestola et al. 2006). The  $K_{70}$  in this study is slightly lower than the  $K_{\tau 0}$  measured by previous authors, but it still follows the trends along the diopside-jadeite join. We note that our best-fit  $K_{T0}$  is significantly smaller than the result from Pandolfo et al. (2012) (5.1  $\pm$  0.3), though the compositions of the two samples are very similar. The potential reasons for this deviation include: (1) The diffractometer in Pandolfo et al. (2012) uses a laboratory MoKa X-ray source, whereas the synchrotron X-ray source used in this study has a higher energy, a better flux and a smaller focus, which help to improve the signal-to-noise ratio. (2) The pressure-transmitting medium used in Pandolfo et al. (2012) is methanol:ethanol:water solution, while the pressure-transmitting medium used in this study is compressed He, which is more hydrostatic and renders higher data quality at high pressures (Dera et al. 2013b).

The relationship between the normalized stress  $(F_{\rm E})$  and the Eulerian strain  $(f_{\rm E})$  gives the visual assessment of the quality of the equation of state fit (Angel 2000). The BM3 equation of state is a weighted linear fit of the  $f_{\rm E}$ - $F_{\rm E}$  dataset (Angel 2000). The  $f_{\rm E}$ - $F_{\rm E}$  plot for omphacite investigated in our study is shown in Fig. 3. The best-fit BM3 equations of state match most of the data, except for a few data points with very small Eulerian strain ( $f_{\rm E} < 0.01$ ), whose uncertainties are large. The  $f_{\rm E}$ - $F_{\rm E}$  datasets with and without the ambient pressure data point have similar levels of scattering. The best-fit BM3 equation of state of the HP only dataset presents a nearly flat line in the  $f_{\rm E}$ - $F_{\rm E}$  plot, because its  $K_{T0}$  is 4.0(2). The HP only dataset can be fitted with a second-order Birch-Murnaghan equation of state, in which the  $V_0$  is 422.7(3) Å<sup>3</sup> and the  $K_{70}$  is 122.2(8) GPa. As we point out in earlier paragraph, the HP only dataset does not necessarily give a better representation of the compressional behavior of this specific omphacite sample, because the accessibility of the available diffraction peaks is limited at



Fig. 4 Pressure dependences of normalized lattice parameters. Weighted BM3 fits of the axial compressional data are plotted in this figure

high pressures, and the ambient pressure data point should give a more accurate unit cell volume than the high-pressure data points.

## Axial compression of P2/n omphacite

The axial compressibilities of minerals are related to the elastic anisotropy. The axial compressibility of an axis ( $\beta$ ) is the inverse of the axial linear modulus (Angel 2000; Hu et al. 2015; Pavese et al. 2001). We use the pressure-length BM3 equations of state to fit the lattice parameters a, b and c (Fig. 4). Then, we calculate their linear moduli from the BM3 equation of state and calculate the axial compressibilities by taking the inverse of the linear moduli. The BM3 fitting is carried out with EoSFit 7c software (Angel et al. 2014), and the fitting was weighted by the uncertainties of both pressure and volume. The best-fit BM3 parameters are listed in Table 1. We notice that the best-fit BM3 parameters of the lattice parameter a are very sensitive to the fitting weight assigned to each data point, yet the bestfit BM3 parameters of b and c are not sensitive to the fitting weight. This is probably because the *a*-axis has the largest error bar among the three axes, which is a result of the limited opening angle of the DAC and the orientation of the crystal. The unweighted best-fit BM3 parameters are listed in Table 1 as references. The pressure dependences of the axial compressibilities are shown in Fig. 5. We notice that the axial compressibility of the *a*-axis is the largest among the three axes at 0 GPa, yet it quickly drops to the smallest at pressures above 5 GPa. Throughout the investigated pressure range, the b-axis is more compressible than the *c*-axis. At 0 GPa, the anisotropy ratio

Table 1Linear BM3fitting parameters and axialcompressibilities of the latticeparameters a, b and c

	a	b	С	
$d_0(\mathbf{A})$	<b>9.585</b> ( <b>4</b> ) 9.57(2)	<b>8.7918</b> (4) 8.79(2)	<b>5.2617</b> ( <b>4</b> ) <i>5.262</i> ( <i>8</i> )	
M <sub>0</sub> (GPa)	<b>244(26)</b> 322(46)	<b>310(4)</b> <i>307(23)</i>	<b>343</b> ( <b>4</b> ) <i>343</i> ( <i>32</i> )	
M'	<b>36(6)</b> 25(5)	<b>8.1</b> (2) 8.2(9)	<b>12.5(3)</b> <i>13</i> (2)	
$\beta_0$ (GPa <sup>-1</sup> )	<b>0.0041(2)</b> 0.0032(5)	<b>0.00324(3)</b> 0.0033(3)	<b>0.00291</b> (2) 0.0029(3)	





Fig. 5 Pressure dependences of the axial compressibilities. Weighted BM3 fits are used to plot this figure

between the three axes using the weighted BM3 fitting is  $\beta_a:\beta_b:\beta_c = 1.41:1.11:1.00$ . The relative ratio of  $\beta_a$  is significantly larger than the results reported by Pandolfo et al. (2012) ( $\beta_a:\beta_b:\beta_c = 1.03:1.07:1.00$ ) and Pavese et al. (2001) ( $\beta_a:\beta_b:\beta_c = 0.95:1.07:1.00$ ). However, when we compare the results using the unweighted best-fit values, the anisotropy ratio changes to  $\beta_a:\beta_b:\beta_c = 1.07:1.12:1.00$ , which is closer to previously reported values. We note here that the *a*- and *c*-axes do not belong to the major compressional axes. The quick change of the axial compressibility of the *a*-axis is explained by the rotation of the strain tensor, which will be discussed in later sections of this manuscript.

In our unit cell setting, the  $\beta$  angle decreases monotonically with pressure (Fig. 6). It decreases from 106.89  $\pm$  0.01° at 0 GPa to 104.30  $\pm$  0.02° at 47 GPa. The decreasing trend of the  $\beta$  angle is consistent with previous studies on *P2/n* omphacite with similar compositions (Pandolfo et al. 2012; Pavese et al. 2001). Similar pressure dependences of the  $\beta$  angle are also observed in other monoclinic pyroxenes, such as diopside (Gavrilenko et al. 2010; Zhang et al. 1997), jadeite (McCarthy et al. 2008; Nestola et al. 2006) and hedenbergite (Hu et al. 2015).



**Fig. 6** Pressure dependences of the  $\beta$  angle of different *P2/n* omphacites

#### **Compression of polyhedra**

We used two parameters to characterize the compression of each polyhedron: the polyhedron volume and the average bond length. The VESTA software was used to calculate these two parameters for each pressure point (Momma and Izumi 2008). We notice that there are some artifacts from the refinement process, which lead to trade-offs in volumes and structures between the M1/M11, M2/M21 and Si1/Si2 polyhedron pairs, so we report the average volumes and average bond lengths of octahedra (M1 and M11), eightcoordinated polyhedra (M2 and M21) and tetrahedra (Si1 and Si2), which show more smooth trends. Among the three categories of polyhedra, the eight-coordinated polyhedra show the fastest compression, while the tetrahedra show the slowest compression. The octahedra show the intermediate compression ratio, and the average octahedron compression ratio is almost identical to the compression ratio of the whole unit cell. We conclude that the compression of the eight-coordinated polyhedra contributes most to the compression of the P2/n omphacite. The best-fit



Fig. 7 Pressure dependences of average polyhedron volumes. *Blue circle* average octahedron volume (M1 and M11). *Green square* average eight-coordinated polyhedron volume (M2 and M21). *Red diamond* average tetrahedron volume (Si1 and Si2). Black dots: unit cell volume. *Blue, red* and *green curves* give the best fits to the BM3 equations of states

 Table 2
 Best-fit BM3 parameters of the average polyhedra volumes as functions of pressure

	Octahedron	Eight-coordinated polyhedron	Tetrahedron
$V_0$ (Å)	10.78(4)	25.48(8)	2.194(5)
$K_{T0}$ (GPa)	118(8)	91(5)	361(5)
K <sub>T</sub> 2032	4.1(5)	3.7(2)	10.0(4)

parameters of the BM3 equations of state of different polyhedra are shown in Fig. 7 and Table 2. The average bond lengths in different polyhedra show the similar behavior as the polyhedron volumes.

#### Strain tensor of omphacite

For a monoclinic mineral whose  $\beta$  angle changes with pressure, the *a*-axis and the *c*-axis of the crystal are not the directions of the minimum and maximum compressions of the crystal structure, i.e., the principle axes of the strain tensor. To fully understand the compression of the crystal under pressures, the analysis of the strain tensor as a function of pressure is needed. We use the Win\_Strain program developed by Dr. Ross Angel to calculate the strain tensor of omphacite at each pressure. The mathematical notations used in the Win\_Strain program are described in an earlier literature (Ohashi 1982). The eigenvalues of the strain



**Fig. 8**  $f_E$ - $F_E$  plot of the three major compression axes. Straight line indicates the weighted linear fit of each  $f_E$ - $F_E$  plot, from which the  $M_0$  and M' are calculated. Top the "soft" major axis in the a-c plane.  $M_0 = 240(16)$  GPa, M' = 21(1). Middle the major axis along the b-axis.  $M_0 = 316(8)$  GPa, M' = 7.9(4). Bottom the "hard" major axis in the a-c plane, which is also the stiffest major axis.  $M_0 = 570(85)$  GPa, M' = 33(5)

tensor and the directions of the major compression axes at each pressure are calculated with the Win\_Strain program. One of the three major compression axes is always parallel to the *b*-axis of the crystal, while the other two axes are within the a-c plane. In the whole pressure range, the stiffest major axis ("hard axis") always lies in the a-cplane. The other major axis ("soft axis") in the a-c plane is the softest axis below 22.9 GPa, yet at pressures above 22.9 GPa, the *b*-axis becomes the softest major axis. The eigenvalues of the strain tensor give the Eulerian finite strain along the three major compression axes. By fitting the f-F plot with a linear function, it is possible to get the isothermal bulk modulus  $(M_0)$  and its pressure derivative (M') of each major compression axis (Fig. 8). With increasing pressure, the orientation of the strain tensor is rotating. To visualize the orientation of the strain tensor, we calculate the representation quadric surface of our sample at each pressure using the methodology described in Knight (2010). One major axis of the representation quadric surface is always parallel to the *b*-axis. In the a-c plane, the long axis of the representation quadric (the stiffest major axis) rotates toward the *a*-axis when the pressure goes higher (Fig. 9). The rotation of the strain tensor explains why the a-axis changes from the most compressible axis to the most incompressible axis (Fig. 5), because the stiffest Fig. 9 Orientations of the representation quadric in the a-c plane for the isothermal compressibility tensor of omphacite at 1.8 and 47.2 GPa, viewed down *b*-axis



major compression axis rotates toward the *a*-axis with the increase in pressure.

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