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## Pressure-induced dehydration of diopside: A single-crystal X-ray diffraction and Raman spectroscopy study

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### ABSTRACT

We present a synchrotron-based, single-crystal X-ray diffraction and Raman spectroscopy study of natural green diopside ( $\text{Cu}_6\text{Si}_6\text{O}_{18}\cdot 6\text{H}_2\text{O}$ ) up to  $\sim 30$  GPa at room temperature. The lattice parameters of diopside exhibit continuous compression behavior up to  $\sim 14.5$  GPa, whereupon a structural transition is observed. Pressure–volume data below 14.5 GPa were fitted to a second-order Birch–Murnaghan equation of state with  $V_0 = 1440(2) \text{ \AA}^3$  and  $K_0 = 107(2) \text{ GPa}$ , with  $K_0' = 4(\text{fixed})$ . The low-pressure form of diopside exhibits anisotropic compression with axial compressibility  $\beta_a > \beta_c$  in a ratio of 1.14:1.00. Based on the diffraction data and Raman spectroscopy, the new high-pressure phase could be regarded as a dehydrated form of diopside in the same symmetry group. Pressure-induced dehydration of diopside contributes broadly to our understanding of the high-pressure crystal chemistry of hydrous silicates containing molecular water groups.

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## 1. Introduction

Silicates comprise more than 90% of the Earth's crust and are thus one of the most fundamental and widespread mineral groups. Pressure-induced transformations of silicate minerals strongly affect the physical properties of the Earth's deep crust and mantle, controlling geological processes and geochemical variations. Hydrous silicates control the recycling of water and of other volatiles in the deep Earth, with implications for volcanism, volatile fluxes, and the buoyancy of subducted slabs

in the Earth's mantle (Faccenda et al., 2012; Mibe et al., 2011; Wallace, 2005). Volatile cycles also impact aspects of planetary dynamics including mantle convection and partial melting.

Cyclosilicates have attracted considerable interest due to their unusual thermal expansion behavior, as well as the ability for the zeolite-like channels to contain a variety of large volatiles as molecular species ( $\text{H}_2\text{O}$  and  $\text{CO}_2$ , etc.). Diopside is a copper cyclosilicate,  $\text{Cu}_6[\text{Si}_6\text{O}_{18}]\cdot 6\text{H}_2\text{O}$ , and crystallizes in a trigonal phase (R and Z = 1) with six-membered  $\text{Si}_6\text{O}_{18}$  rings. Ribbe et al. (1977) described the structure as consisting of distorted trigonal rings of six water molecules in an ice-like configuration between similarly puckered trigonal rings of silicate tetrahedra coordinated laterally and vertically by Cu atoms (Frost and Xi, 2013). Distorted  $\text{CuO}_4(\text{H}_2\text{O})_2$  octahedra share edges to

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form dimers and link the silicate rings into a three-dimensional framework (Fig. 1).

The crystal structure of diopside was refined by Ribbe et al. (1977), followed by a detailed charge density study by Belokoneva et al. (2002). Breuer et al. (1989) and Breuer and Eysel (1988) found two high-temperature polymorphs, the first on heating to above 400 °C was called blue diopside, and the second on further heating led to a dehydrated black diopside,  $\text{Cu}_6\text{Si}_6\text{O}_{18}$ , with closely related structure. The vibrational states of diopside during the dehydration process have also been investigated using IR spectroscopy by Goryainov (1996). Frost and Xi (2013) studied the thermal decomposition of diopside up to 1000 °C, resulting in a wide temperature range for loss of water (400–730 °C) and two mass-loss stages for losing oxygen (~793 and 835 °C). Recently, diopside has attracted attention as an antiferromagnetic material that exhibits strong quantum fluctuations in the system and reduced Néel temperature,  $T_N = 14.5$  K (Janson et al., 2010; Podlesnyak et al., 2016). Wang et al. (2015) studied the compressional behavior of single-crystal diopside up to 11.7 GPa, resulting in  $K_T = 107(6)$  GPa and  $K'_T = 3.4(1.2)$ . Whereas most studies of diopside have focused on the high-temperature decomposition of green diopside and related magnetic properties, the elastic behavior and potential pressure-induced structural transformations require further study.

We investigated the compression behaviour of diopside up to ~30 GPa at room temperature in two separate diamond-anvil cell runs. A pressure-induced phase transition was observed by X-ray diffraction. High-pressure Raman spectroscopy experiment suggests that the transition is associated with dehydration. Knowledge of high-pressure diopside crystal chemistry contributes broadly to our understanding of how silicate minerals containing water molecules behave under compression.

## 2. Experimental methods

Single-crystal samples of natural, gem-quality green diopside were used for high-pressure diffraction and Raman spectroscopy experiments. The diopside sample

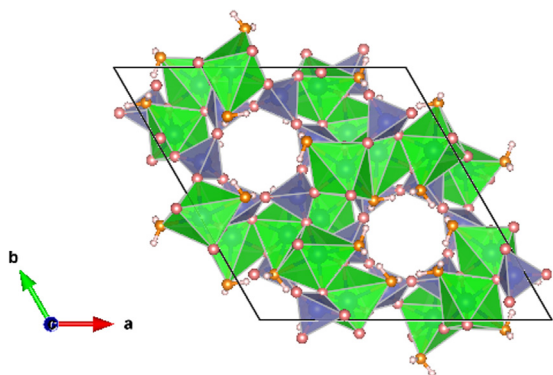


Fig. 1. Crystal structure of diopside. Blue-colored polyhedra represent  $\text{SiO}_4$  tetrahedra. Green-colored polyhedra represent  $\text{CuO}_4(\text{H}_2\text{O})_2$  octahedra. The orange-colored spheres are oxygen atoms of the  $\text{H}_2\text{O}$  molecules and grey-colored spheres represent oxygen atoms.

measured in this study has the same chemical composition as that of in Ruan et al. (2016), including 45.48%  $\text{CuO}$  and 42.33%  $\text{SiO}_2$  as well as some minor components  $\text{MgO}$  (0.23%),  $\text{FeO}$  (0.01%) and  $\text{CaO}$  (0.05%). The ratio of Cu to Si was estimated to be 1:1.08 using an energy-dispersive detector on the FEI Quanta 650 scanning electron microscope with an accelerating voltage of 15 kV and a beam current of 20 nA at Peking University. The results suggested that the chemical formula of the diopside sample is close to  $\text{Cu}_6[\text{Si}_6\text{O}_{18}]\cdot 6\text{H}_2\text{O}$ .

A symmetric-type diamond anvil cell (DAC) and a short symmetric-type DAC fitted with Boehler–Almax diamond anvils were employed to approach the high-pressure conditions in both X-ray diffraction (XRD) and Raman spectroscopic studies. Rhenium gaskets were pre-indented by diamond anvils having 300  $\mu\text{m}$  flat culets to a thickness of ~35  $\mu\text{m}$ . Holes were drilled to ~170  $\mu\text{m}$  in diameter and used as the chamber chambers. Single-crystal samples were polished on both sides down to ~10  $\mu\text{m}$  thickness using 3 M™ diamond lapping films. A Pt foil was used for pressure calibration in XRD experiments, and a small Ruby sphere was for use as a pressure calibrant in Raman spectroscopic study (Fei et al., 2007; Mao et al., 1986). To achieve quasi-hydrostatic conditions, the cells were loaded with neon as the pressure-transmitting medium using the COMPRES/GSECARS gas-loading system (Rivers et al., 2008).

In run 1, high-pressure, single-crystal X-ray diffraction experiments were conducted at Beamline 13-BM-C of GSECARS (Sector 13) of the Advanced Photon Source (APS), Argonne National Laboratory (ANL). A monochromatic X-ray beam with wavelength 0.4340 Å was focused to a  $15 \times 15 \mu\text{m}^2$  spot, and the experimental details were described previously (Qin et al., 2017; Zhang et al., 2017). To obtain the adequate number of diffraction peaks of samples and increase the coverage of the reciprocal space, we collected data at four different detector positions. The diffraction images were analyzed using the ATREX/RSV software package (Dera et al., 2013).

In run 2, *in-situ* high-pressure single-crystal X-ray diffraction experiments on Beamline 13-ID-D used a monochromatic X-ray beam with a wavelength of 0.3344 Å and tightly focused to a  $5 \times 5 \mu\text{m}^2$  spot. XRD patterns were recorded with a MARCCD detector, and the exposure time was 10 s for each crystal diffraction pattern. DAC was continuously rotated from  $-10^\circ$  to  $10^\circ$  (wide-scan mode) along the X-ray beam direction. The  $d$  spacings of independent diffraction peaks in single-crystal diopside at various pressures were extracted by using the ATREX/RSV software package (Dera et al., 2013), and then the unit-cell parameters were refined using UnitCell software (Holland and Redfern, 1997). In order to maintain consistency, only the peaks appearing at all pressure steps were included for high-pressure lattice parameter refinements.

A fragment of the diopside sample used in the XRD studies was also investigated with confocal micro-Raman spectroscopy at Northwestern University. High-pressure Raman spectroscopy was performed in a symmetric DAC with 300  $\mu\text{m}$  culet ultralow fluorescence diamonds in a pressure medium of neon up to 22.5 GPa at room

temperature. All analyses were performed in backscattering geometry using an Andor Shamrock 0.3 m spectrograph (1800 grooves/mm grating) coupled with an Andor Newton DU970 EM-CCD camera, an Olympus optical microscope and a long-working-distance Mitutoyo 20× objective. The 458 nm line of a solid-state Melles Griot laser source with ~300 mW output power was used for sample excitation. Laser intensity at the sample was ~18 mW. Counting times were 20 s at 6 accumulations. The Raman peak positions were fitted using the software package PeakFit (Systat Software, Inc.).

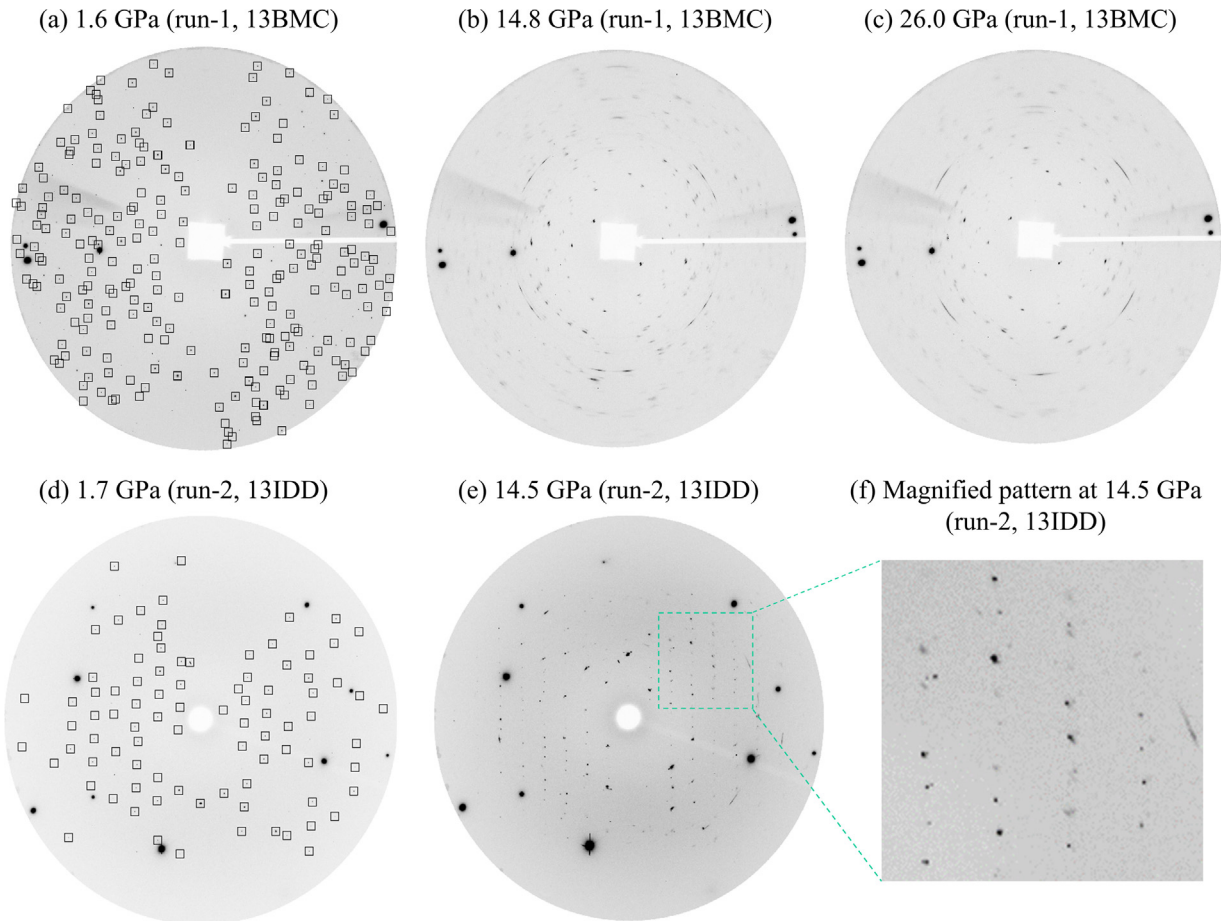
### 3. Results and discussion

#### 3.1. Structural variations of diopside

In run 1, high-pressure single-crystal XRD data for natural diopside were measured up to 26.0 GPa at room temperature. Fig. 2 (a, b and c) shows the single-crystal diffraction patterns at three selected pressures during compression (1.6, 14.8, and 26.0 GPa, respectively). Reflections from the diopside crystal initially appear as sharp and round spots in the patterns and can be well

indexed to the *R* structure. At 14.8 GPa, an abrupt change in the diffraction pattern was observed. Reflections from the crystal show diffuse scattering and appear as short streaks at pressures above 14.8 GPa. At higher angle, diffraction peaks become fainter with increasing pressure, making it difficult to determine the peak positions exactly (Fig. 2c). In run 2, high-pressure synchrotron XRD patterns for diopside were collected up to 30.6 GPa at 300 K. Single-crystal diffraction patterns taken at 1.7 and 14.5 GPa in wide-scan mode are shown in Fig. 2d and 2e. The compression in run 2 duplicates the structural change observed in run 1. At 14.5 GPa, some new peaks become visible, adjacent to the main diffraction peaks (Fig. 2f). The *d* values of the new peaks are different from the characteristic peaks of diopside, indicating that a structural modification of diopside occurred.

The integrated XRD patterns of diopside in both runs are presented in Fig. 3. At the first pressure point, the diffraction peaks were sharp and distinct, and the volume can be comparable with those obtained at ambient conditions reported by Belov et al. (1978) using X-ray and neutron diffraction methods. With increasing pressure, all the diffraction peaks shift to larger  $2\theta$  values, as



**Fig. 2.** Diffraction patterns of diopside at selected pressures. Boxes are shown around the diffraction peaks that were used in the analysis. The expanded image of diopside diffraction peaks at 14.5 GPa shown in (f) demonstrates the appearance of new peaks and the formation of a new phase. The very intense spots are from the diamond anvils. Diamond peaks and diffraction lines attributed to neon are not marked.

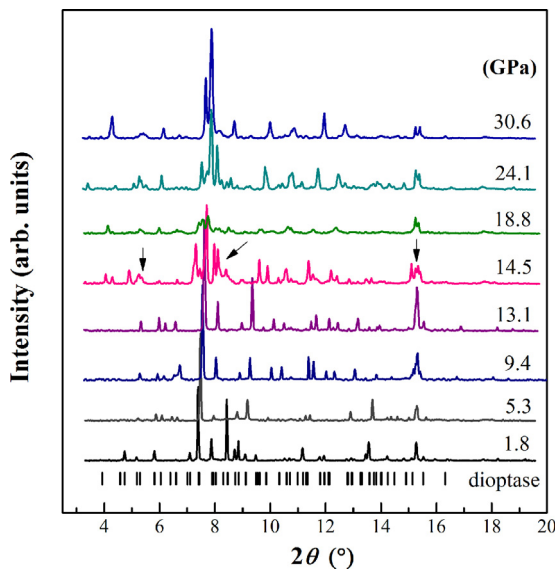


Fig. 3. Integrated XRD patterns of diopside at selected pressures in run 2. Backgrounds were subtracted from the original data. The calculated peak positions of diopside are shown by small ticks.

expected, and their peak intensities gradually weaken (Fig. 3). When the pressure exceeds  $\sim 14$  GPa, the most notable feature is the appearance of peak overlap and twin peaks at about  $5.5^\circ$ ,  $7.5^\circ$ , and  $15.0^\circ$  (indicated by small arrow in Fig. 3), corresponding to the two-dimensional XRD patterns. The new peaks correspond to the dehydrated phase, black diopside,  $\text{Cu}_6\text{Si}_6\text{O}_{18}$  (Breuer et al., 1989), thus we infer that green diopside undergoes a pressure-induced dehydration phase transition at  $\sim 14.5$  GPa. With further compression to 30 GPa, no other phase transition was observed.

### 3.2. Elastic behavior of diopside

Lattice parameters and volume compression data for green diopside from the two separate runs are combined in Fig. 4 and Table 1. The volume-pressure dataset up to 14.0 GPa were fitted to a third-order Birch–Murnaghan equation of state (BM3-EoS) using error-weighted least squares with EoSFit7c (Angel et al., 2014). The resulting BM3-EoS parameters are as follows:  $V_0 = 1441(3) \text{ \AA}^3$ ,  $K_0 = 103(9) \text{ GPa}$  and  $K_0' = 4(1)$ , which is about 0.8% larger than volume at ambient conditions ( $1429.9(5) \text{ \AA}^3$ ) measured by Belov et al. (1978) using X-ray and neutron diffraction methods, the difference in  $K_0$  is mainly due to the difference in experimental methods. We next performed BM2-EoS ( $K_0' = 4$  implied) fit to the current dataset and yielding  $V_0 = 1440(2) \text{ \AA}^3$  and  $K_0 = 107(2) \text{ GPa}$ . However, we could not determine the exact lattice constant of high-pressure phase diopside above 14.5 GPa due to the limitations in quality of the diffraction peaks and limited angular access of the DAC. Furthermore, because of the structural similarity between green and black diopside (dehydrated form), we could discern no significant difference in lattice constants.

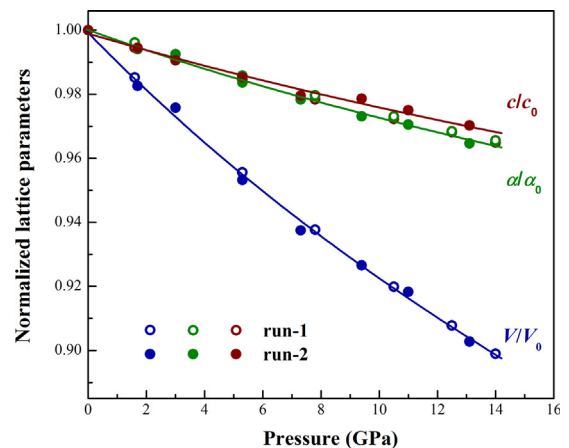


Fig. 4. Normalized lattice parameters of diopside as a function of pressure at room temperature. Second-order Birch–Murnaghan fits to the volume compression are shown by solid lines.

Axial compression data of green diopside are plotted in Fig. 4. The axial compressibility of  $a$  and  $c$  in diopside were fitted to a modified BM2-EoS, where in each axial dimension is cubed and treated as volume in the BM formulation (Angel et al., 2014; Xia et al., 1998). The zero-pressure axial compressibility of linear dimension  $l$  is defined as  $\beta_{l0} = -(l^{-1})(\delta l/\delta P)_{P=0}$ , which is related to the linear modulus (linear incompressibility) by  $M_{l0} = (\beta_{l0})^{-1}$ . Our fitted linear moduli to  $a$  and  $c$  are  $306(11)$  and  $340(19) \text{ GPa}$  with an implied value for  $K_0'$  of 12, respectively, corresponding to axial compressibility values of  $\beta_a = 3.3(2) \times 10^{-3}$  and  $\beta_c = 2.9(2) \times 10^{-3} \text{ GPa}^{-1}$  for diopside. We find that the green diopside exhibits anisotropy in axial compressibility with the  $a$ -axis appears more compressible than the  $c$ -axis (Fig. 4).

### 3.3. Raman spectroscopy

Raman spectroscopic data were obtained for the same natural diopside sample up to a maximum pressure of 22.5 GPa. Raman spectra in the low-frequency ( $100\text{--}1100 \text{ cm}^{-1}$ ) and high-frequency ( $3000\text{--}3700 \text{ cm}^{-1}$ ) ranges are displayed in Fig. 5. Group theory analysis of green diopside demonstrates that the total irreducible representation is  $\Gamma = 18A_g + 18^1E_g + 18^2E_g$  (Aroyo et al., 2006a, b). However, only a small number of active Raman modes could be observed due to the weak intensity, peak overlap and orientation dependence.

At 0.7 GPa, the Raman spectra are similar to those in the University of Arizona RRUFF database (Lafuente et al., 2015) #X050061. In the low-frequency region, all Raman modes show a continuous shift to higher frequency with increasing pressure (Fig. 5). Most of the modes resolved in this study are assigned to Si–O stretching or bending vibrations or ring distortions. The intense bands in the range of  $200\text{--}500 \text{ cm}^{-1}$  are assigned to the external vibrations of  $\text{SiO}_4$ . Two strong peaks are located at  $527$  and  $665 \text{ cm}^{-1}$ , which are attributed to the O–Si–O internal vibrations of  $\text{SiO}_4$ . Four Raman modes observed at  $748$ ,  $921$ ,  $974$ ,  $1012 \text{ cm}^{-1}$  empirically arise from symmetric

**Table 1**  
Lattice parameters of diopside at various pressures.

Pressure (GPa)	<i>a</i> (Å)	<i>c</i> (Å)	<i>V</i> (Å <sup>3</sup> )
13BMC			
1.6	14.532(1)	7.7649(7)	1420.1(2)
5.3	14.387(1)	7.6837(7)	1377.4(2)
7.8	14.296(1)	7.6363(6)	1351.6(2)
10.5	14.207(1)	7.5855(6)	1326.9(2)
12.5	14.147(2)	7.549(2)	1308.4(4)
14.0	14.099(1)	7.527(1)	1295.8(2)
13IDD			
1.7	14.5246(6)	7.7521(2)	1416.3(1)
3.2	14.5024(9)	7.7217(3)	1406.4(1)
3.7	14.4919(9)	7.7119(2)	1402.6(1)
5.3	14.3729(8)	7.6799(2)	1373.9(1)
7.3	14.2951(9)	7.6353(3)	1351.2(1)
9.4	14.2187(8)	7.6283(2)	1335.6(1)
11.0	14.1807(8)	7.6003(2)	1323.6(1)
13.1	14.0955(8)	7.5628(2)	1301.3(1)

stretching of Si–O. Notably, the vibration modes at 137, 165 and 181 cm<sup>-1</sup> are assigned to Cu–O vibrations. The positions of these observed bands agree well with the infrared data reported by Goryainov (1996). Raman mode assignments are summarized in Table 2.

In the high-frequency range from 3000–3700 cm<sup>-1</sup>, there is an intense peak at 3372 cm<sup>-1</sup> (FWHM: 58 cm<sup>-1</sup>), which is associated with the O–H stretching vibration of molecular H<sub>2</sub>O in the structure (Fig. 5). Previous infrared studies of diopside reported the O–H band at ~3363 cm<sup>-1</sup> (Frost and Xi, 2013). Goryainov (1996) collected high-temperature infrared spectra, finding the O–H stretching vibration at 3420 cm<sup>-1</sup> at 500–600 °C. In the current study, we observe the 3372 cm<sup>-1</sup> band shifts to lower frequency

with increasing pressure at a rate ~4.3 cm<sup>-1</sup>/GPa and disappears abruptly between 12.8 and 16.9 GPa, indicating loss of H<sub>2</sub>O molecules in the structure.

With increasing pressure, the low-frequency Raman modes shift to higher frequencies and generally decrease in intensity (Fig. 5). Below ~12 GPa, low-wavenumber region Raman modes remain sharp, except for the 665 and 750 cm<sup>-1</sup> modes, which broaden significantly. Notably, the intense mode at ~360 cm<sup>-1</sup> has a slightly positive shift at a rate ~0.2 cm<sup>-1</sup>/GPa under compression. The Raman spectra are significantly different at 16.8 GPa and higher, where many of the sharp modes have disappeared and the appearance of a new intense band at 320 cm<sup>-1</sup> is observed. Between 16.8 and 22.5 GPa, the Raman spectra do not significantly change. Raman mode frequencies are plotted as a function of pressure in Fig. 6.

#### 4. Discussion

The high-pressure behavior of ring silicates has attracted interest due to their occurrence in subduction zone environments as well as their potential applications in having channeled structures. The compressibility of diopside is compared with other hydrous cyclosilicates [Si<sub>6</sub>O<sub>18</sub>] in Fig. 7, including beryl (Al<sub>2</sub>Be<sub>3</sub>Si<sub>6</sub>O<sub>18</sub>), cordierite (Mg<sub>2</sub>Al<sub>3</sub>(AlSi<sub>5</sub>)O<sub>18</sub>), diopside (Cu<sub>6</sub>[Si<sub>6</sub>O<sub>18</sub>]·6H<sub>2</sub>O) and tourmaline (NaMg<sub>3</sub>Al<sub>6</sub>Si<sub>6</sub>O<sub>18</sub>(BO<sub>3</sub>)<sub>3</sub>(OH)<sub>3</sub>F) (Fan et al., 2015; Miletich et al., 2014; Wang et al., 2015; Xu et al., 2016). Referring to Fig. 7, reported values of the bulk modulus *K*<sub>0</sub> and pressure derivative *K*<sub>0</sub>' of hydrous cyclosilicates range significantly, from 99 to 173 GPa for *K*<sub>0</sub> and from 0.3 to 12 for *K*<sub>0</sub>'. The pressure derivative *K*<sub>0</sub>' of diopside found in this study is very close to 4 using the BM3-EoS fit, which is

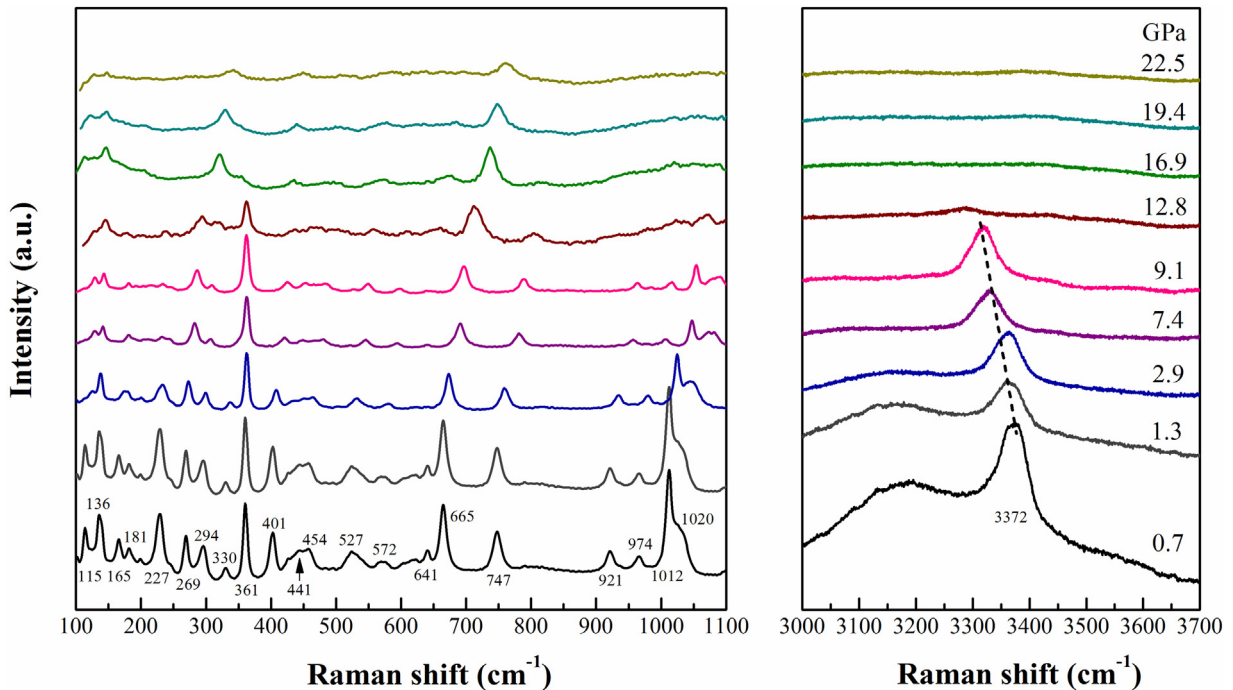
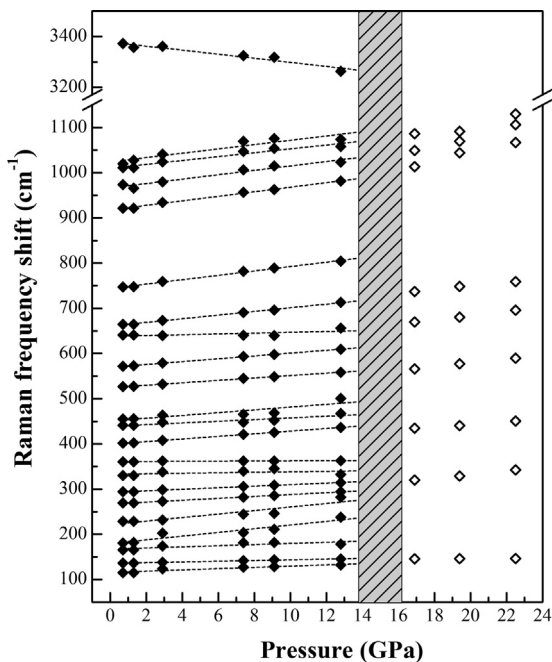


Fig. 5. Selected Raman spectra of green diopside at various pressures.

**Table 2**  
Results of linear regressions to the pressure dependence of observed vibrational modes ( $\nu_i$ ) in diopside (green).

Modes	$\omega_0$ (cm <sup>-1</sup> )	$\alpha$ (cm <sup>-1</sup> /GPa)	$\gamma_i$	Assignment
$\nu_1$	116	1.4(2)	1.29(2)	
$\nu_2$	136	0.78(5)	0.61(4)	Cu–O
$\nu_3$	167	1.3(4)	0.83(3)	Cu–O
$\nu_4$	180	4.1(7)	2.44(4)	Cu–O
$\nu_5$	222	3.9(8)	1.88(4)	Si–O
$\nu_6$	267	2.09(4)	0.84(3)	Si–O
$\nu_7$	293	1.70(5)	0.62(3)	Si–O
$\nu_8$	333	0.54(6)	0.17(3)	Si–O
$\nu_9$	361	0.17(7)	0.05(1)	Si–O
$\nu_{10}$	399	2.90(5)	0.78(1)	Si–O
$\nu_{11}$	439	1.9(4)	0.46(9)	
$\nu_{12}$	451	3.1(8)	0.74(9)	Si–O
$\nu_{13}$	525	2.67(7)	0.54(4)	Si–O
$\nu_{14}$	570	3.13(6)	0.59(3)	Si–O
$\nu_{15}$	638	0.9(5)	0.15(1)	Si–O
$\nu_{16}$	661	4.0(1)	0.65(1)	Si–O
$\nu_{17}$	744	4.9(2)	0.70(1)	Si–O
$\nu_{18}$	918	5.0(2)	0.58(1)	Si–O
$\nu_{19}$	966	4.9(5)	0.54(1)	Si–O
$\nu_{20}$	1010	4.3(5)	0.46(1)	Si–O
$\nu_{21}$	1024	4.8(9)	0.50(1)	Si–O
$\nu_{22}$	3378	-8.0(10)	-0.25(2)	OH

a typical value for silicates. There is a larger difference in  $K_0$  and  $K_0'$  between diopside and other cyclosilicates, possibly due to the different incorporation mechanisms of H<sub>2</sub>O (e.g., H<sub>2</sub>O in diopside and beryl, OH<sup>-</sup> in tourmaline) as well as different sizes of metal ions (CN = 6, Cu<sup>2+</sup>: 0.73 Å, Al<sup>3+</sup>: 0.535 Å and Mg<sup>2+</sup>: 0.72 Å) (Shannon, 1976). The ratio of zero-pressure axial compressibility in diopside is 1.14:1.00



**Fig. 6.** Pressure dependence of the observed vibrational frequency modes for diopside. The grey-shaded region indicates the possible phase boundary between diopside (green) and dehydrated diopside (black). Solid lines were obtained using linear regressions, provided in Table 2.

( $\beta_a$ :  $\beta_c$ ), which can be explained by the three-dimensional configuration of Si<sub>6</sub>O<sub>18</sub> rings, Cu octahedra, and water molecules. The Si<sub>6</sub>O<sub>18</sub> silicates and H<sub>2</sub>O rings form a vertical stack parallel to the *c*-axis, leading to a denser structure along the *c*-axis. In addition, the slightly higher compressibility along the *a*-axis may be caused by the configuration of water molecules and larger Cu<sup>2+</sup> ions along the *a*-axis.

Raman spectroscopy is a sensitive technique to probe the characteristics of cation–anion coordination groups. The pressure coefficients give quantitative comparisons for the changes of different vibration modes with pressure. Raman modes are assigned by comparing our results with previous studies, and the results are summarized in Table 2 (Goryainov, 1996). We observe about 22 vibrational mode frequencies ( $\nu_i$ ), compared with group theory calculations of 54 Raman-active modes. Mode frequencies were fitted linearly by  $\omega_i = \omega_{i0} + \alpha_i \times P$ , where  $\omega_{i0}$  is the frequency of mode *i* at 0 GPa,  $\alpha_i$  is the linear pressure coefficient ( $\partial\omega_i/\partial P$ )<sub>*P*=0</sub>, and *P* is the pressure. Thus, the mode Grüneisen parameters ( $\gamma_i$ ) can be calculated using  $\gamma_{i0} = -\partial(\ln\omega_i)/\partial(\ln V) = (K_{T0}/\omega_{i0})\alpha_i$  (Born and Huang, 1954). Here,  $K_0$  is the bulk modulus of diopside, for which we used a value of 107(2) GPa from this study. The pressure coefficient values of the Cu–O bonds fall generally in the range from 0.78(5) to 4.1(7), which is attributed to the different vibration modes in the Cu<sub>2</sub>O<sub>4</sub> polyhedron. The resulting average mode Grüneisen parameter of Cu–O bond is 1.29. Notably, as for the silicate-derived vibration in diopside, the calculated  $\gamma$  is 0.61, which is a typical value for some hydrous silicates. The overall average mode Grüneisen parameter for the Raman bands is 0.67 (Table 2).

The single-crystal X-ray diffraction study of diopside provides information on the evolution from green diopside to its high-pressure phase (black or dehydrated diopside). The transition pressure in this study was determined to be  $14.5 \pm 0.3$  GPa and confirmed in two runs. Previous studies have found that natural green diopside transforms to the black-colored (dehydrated) diopside Cu<sub>6</sub>[Si<sub>6</sub>O<sub>18</sub>] between 400 and 800 °C (Breuer and Eysel, 1988; Breuer et al., 1989). In the present study, combined Raman spectroscopy and XRD data are used to infer a similar structural transition at high pressure. Additionally, Frost and Xi (2013) noted a two-stage mechanism for diopside dehydration, which transformed into the blue polymorph first and then dehydrated to a black diopside Cu<sub>6</sub>[Si<sub>6</sub>O<sub>18</sub>]. In this study, we found that the green diopside transformed to the black diopside at around 14.5 GPa and a mixed-phase region of green and black diopside from 14.5 to 30.6 GPa was observed from the XRD results. However, we are not able to determine where and how hydrogen is located from the current methods, so the phase boundary between green, blue, and black diopside is still unknown and requires further study.

The pressure-induced dehydration of diopside likely contributes to the slightly unbending of Si<sub>6</sub>O<sub>18</sub> ring silicate and shorter Si–O bonds. The Cu–O distance could also be reduced, drawing the silicate rings together and closing the channels for water molecules (Goryainov, 1996). The evidence from density functional theory calculation also proved that the effect of pressure may lead to substantial reorganizations of the O–H network and modifications in

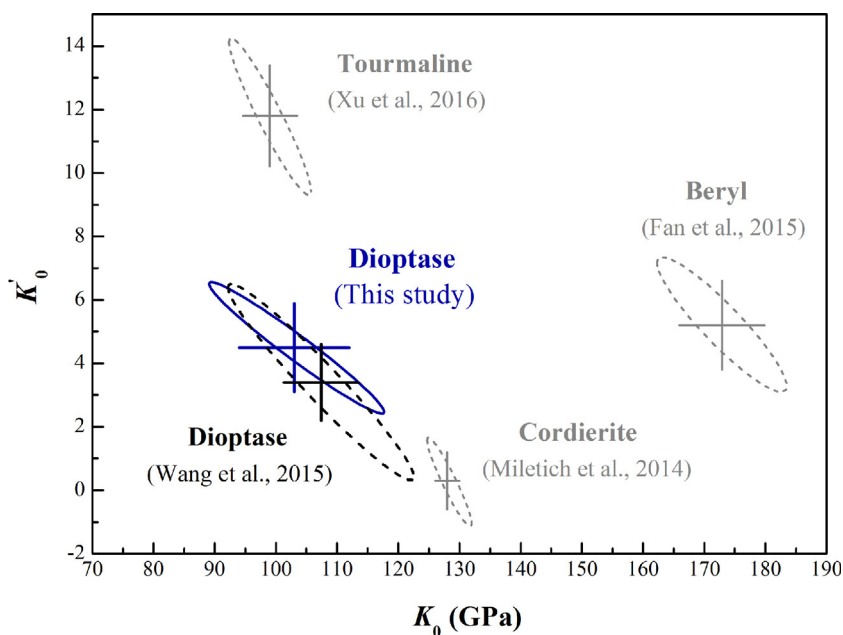


Fig. 7. Isothermal bulk moduli and their pressure derivatives of some cyclosilicates, plotted as confidence ellipsoids at the 95.4% level.

the coordination environment as well (Brand et al., 2009). Similarly, it is expected that some hydrous minerals, may process the same appearance of dehydration at elevated pressures, such as kaersutite and gypsum (Comodi et al., 2012; Zhu et al., 2004). Since this phenomenon was induced by the departure of crystallization water in diopase, and the probable removing water molecules may still exist in the sample chamber. Moreover, green-colored diopase and black diopase have attracted wide attention from the interplay between its crystallographic and magnetic properties. Two magnetic modes and prominent spin gap in green diopase are consistent with the ground state of Cu moments coupled antiferromagnetically in spiral chains along the *c*-axis (Matsui et al., 2014; Podlesnyak et al., 2016). It can be expected that the transition from green diopase to dehydrated black diopase at high pressure also changes the magnetic properties of diopase. Further studies of diopase at high pressure are required to elucidate the pressure-induced dehydration mechanism and the black-diopase structure at high pressure, as well as the associated physical properties.

## 5. Conclusions

In summary, a pressure-induced dehydration study of hydrous cyclosilicate diopase was carried out using synchrotron-based XRD and Raman spectroscopy at pressures up to ~30 GPa. The trigonal diopase phase exhibits anisotropic compression with  $\beta_a > \beta_c$ . The overall average mode Grüneisen parameter for the Raman bands is 0.67. A high-pressure phase of diopase was observed at around ~14.5 GPa. From Raman spectroscopy, we infer that the structural variation is associated with a pressure-induced dehydration of diopase to black diopase,

$\text{Cu}_6\text{Si}_6\text{O}_{18}$ . This study contributes to broadening our understanding of the high-pressure crystal chemistry of hydrous silicates containing water molecules.

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